



計畫名稱：具不同立體異構性 PMAs 和另一高分子摻和研究

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中文摘要：

亂排 PMMA 和同排 PMMA 及 PSAN 都相容，但同排 PMMA 和 PSAN 卻不相容。本研究主要在探討同排 PMMA、亂排 PMMA 和 PSAN 三種混合物的相容性並建立它們的三相圖。實驗結果顯示當三相混合物含高濃度同排 PMMA 及低濃度 PSAN 時呈現不相容性。

ABSTRACT

Previously, atactic and syndiotactic poly (methyl methacrylates) (PMMA) were found to be miscible with poly (styrene-co-acrylonitrile) because all the prepared films were transparent and showed composition dependent glass transition temperatures (T_g s). However, the isotactic PMMA/PSAN blend was immiscible because most of the cast films were translucent and had two glass transition temperatures. Isotactic PMMA is known to be miscible with atactic PMMA according to literature. Atactic PMMA is miscible with both PSAN and isotactic PMMA but PSAN and isotactic PMMA are immiscible. It will be interesting to study the miscibility of ternary blends of isotactic and atactic PMMA and PSAN. An investigation based on this is reported in this article. Calorimetry was used as the principal tool to study miscibility. A blend composed of isotactic and atactic PMMA was also prepared and confirmed to be miscible. T_g of this blend seemed to vary linearly with composition. An approximate phase diagram of the ternary blends was established based on differential scanning calorimetry data. The results indicated that the ternary blends rich in isotactic PMMA and with a low PSAN concentration were found to be immiscible. One possible cause is attributed to more repulsive MMA-styrene interaction and/or MMA-AN interaction between isotactic PMMA and PSAN.

Key words: tacticity/ PMMA/ PSAN/ternary blends

INTRODUCTION

Ternary blends are gaining importance in the field of polymers. The first systematic study on ternary blends was reported by Kwei *et al.*¹ in 1977. In their study, the addition of poly(vinylidene fluoride) (PVDF) to the immiscible pair poly(methyl methacrylate) (PMMA)/poly(ethyl methacrylate) (PEMA) was studied and found to be miscible. A list of ternaries investigated has been considerably enlarged.²⁻⁸ In nearly all these blends, a third component either a homopolymer or copolymer is added to homogenize an immiscible pair. Miscibility is often achieved in cases where this third component is miscible with other polymers.

Recently, Bicakci and Cakmak⁹ investigated the phase behavior of binary and ternary blends of poly(ethylene naphthalate) (PEN), poly(ether imide) (PEI) and poly(ether ether ketone) (PEEK) using differential scanning calorimetry (DSC) and dynamic mechanical analysis (DMTA) techniques. PEN/PEI and PEI/PEEK binary blends exhibit single glass transition temperatures in full composition range and PEN and PEEK were found immiscible particularly at mid-concentration. When PEI is added to the immiscible PEN/PEEK system, first the blends form two separated PEN-rich and PEEK-rich phases below about 40% PEI concentration, and above this concentration, the three homopolymers form a

miscible phase in the amorphous state exhibiting a single T_g . An approximate ternary phase diagram was established based on the DSC and DMTA results.

In a previous study,¹⁰ isotactic, atactic, and syndiotactic PMMA (designated as i, a, and s-PMMA) with approximately the same molecular weight were blended with poly(styrene-co-acrylonitrile) (PSAN) (containing 25 wt% of acrylonitrile) in tetrahydrofuran to cast into films. The glass transition temperatures of the polymers were measured. aPMMA and sPMMA were found miscible with PSAN because all the prepared films were transparent and had a single composition dependent glass transition temperature (T_g). However, iPMMMA is not miscible with PSAN because most of the cast films were translucent and had two T_g s. Since iPMMMA is miscible with aPMMA according to the literature¹¹ but immiscible with PSAN and aPMMA is miscible with PSAN. Therefore an investigation of the miscibility of ternary blends composed of iPMMMA, aPMMA, and PSAN is worthwhile and was pursued in this laboratory. aPMMA is acting as a "cosolvent" between PSAN and iPMMMA in this study.

In this article, the ternary blends of iPMMMA, aPMMA, and PSAN were prepared in several weight ratios. The glass transition temperatures of the ternary mixtures were determined calorimetrically. A blend of iPMMMA and aPMMA was made to confirm miscibility. An approximate phase diagram of the ternary blends was established based on calorimetry data and a single T_g was used as the criterion for determining miscibility. The results indicated that the formation of miscible ternary blends is favored when there is a high content of PSAN and/or aPMMA in the blends.

EXPERIMENTAL

Materials

Isotactic and atactic PMMA (designated as i and a-PMMA in this study) were purchased from Polysciences, Inc, Warrington, PA. According to supplier information, the molecular weights (M_w s) of iPMMMA and aPMMA are the same about 100000. The PSAN contained 25 wt% AN units and was purchased from Aldrich Chemical Company Inc., Milwaukee, WI. The M_w value for PSAN is 165000. iPMMMA was mixed with aPMMA to form a binary blend in the weight ratios of 1/3, 1/1, and 3/1. A ternary mixture of iPMMMA, aPMMA, and PSAN in different weight ratios was also prepared. The actual compositions of the binary and ternary blends are shown in Tables I and II, respectively.

Film Preparation

Thin films of binary and ternary blends were made by solution casting onto glass plates. Tetrahydrofuran (THF) was used as solvent for most compositions, but for PMMA toluene was used instead. THF and toluene are A.C.S. reagent purchased from Fisher Scientific, Fair Lawn, New Jersey. The final drying step for all the films took place in a vacuum oven for about one day at 92–125°C, which was above the glass transition temperatures of the individual polymers. The films were cooled to room temperature slowly by air cooling to make as-cast samples. The as-cast samples were used for DSC studies.

Differential Scanning Calorimetry (DSC)

Glass transition temperatures (T_g s) of the polymer blends were determined by a DuPont 2000 thermal analyzer. The experiments were performed in two consecutive scans in the ambient environment of nitrogen gas at a flowing rate of 100–110 ml min⁻¹. At the end of the first thermal scan, the samples stayed at 200°C for 1 min. The samples were quenched to 0°C immediately using an ice-water bath and were scanned the second time. The scanning temperature from 30 to 200°C and heating rate of 20°C min⁻¹ were used in each scan. The samples were quenched quickly to prevent crystallization as detected by DSC. There was no trace of solvent in the films detectable by DSC. The inflection point of the specific heat jump of a thermal scan was taken as the glass transition temperature. The glass transition temperatures determined from the first and second thermal scans were designated as T_{gsc} and T_{gq} , respectively. Therefore, T_{gsc} is the T_g of the slowly cooled (as-cast) films and T_{gq} that of the quenched films.

RESULTS AND DISCUSSION

We did not characterize the tacticity of PMMA by NMR. A simple estimation of the *meso* and *racemic* diad fractions of PMMA was resorted to. The *meso* and *racemic* diad fractions of PMMA were estimated previously^{1,2} according to Fernandez-Martin *et al.*^{1,3} or Allen *et al.*^{1,1} The average *meso* fractions from these two methods are 0.687, 0.338, and 0.093 for iPMMA, aPMMA, and sPMMA, respectively.

Fox and Weight Average Equations

For polymer blends with weak or no interaction, the Fox equation¹⁴ seems to predict the glass transition temperature quite well. The Fox equation extended for a ternary mixture is shown

$$1/T_g = w_1/T_{g1} + w_2/T_{g2} + w_3/T_{g3} \quad (1)$$

where T_g is the glass transition temperature of a blend. w_1 , w_2 , and w_3 are the weight fractions of polymers 1, 2, and 3, respectively. For a binary mixture, $w_3 = 0$.

The weight average equation for a ternary blend is presented below as in eq 2

$$T_g = w_1 T_{g1} + w_2 T_{g2} + w_3 T_{g3} \quad (2)$$

where symbols have the same meanings as in eq 1 and again $w_3 = 0$ for a binary one. The Fox and weight average equations were used to estimate T_g of the binary and ternary blends.

Binary Blend of iPMMA and aPMMA

The iPMMA/aPMMA blend after preparation appeared transparent indicating possible miscibility. The second thermal scans from 30°C to 180°C of the same blend are presented in Figure 1. Estimated T_{gsc} and T_{gq} of the iPMMA/aPMMA blend are listed in Table I. There is not much difference between most T_{gsc} and T_{gq} except for one composition. For iPMMA/aPMMA (75.0/25.0),

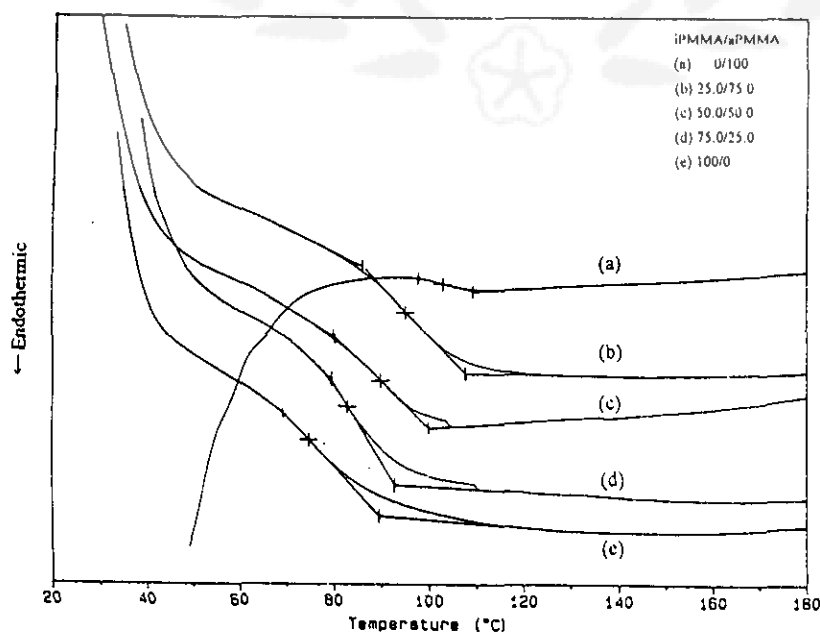


Figure 1. DSC thermograms of the iPMMA/aPMMA blend.

the as-cast film had two T_g 's but the quenched sample had one T_g . Because T_{gq} are considered to be free of annealing or other thermal/solvent effects, the iPMMA/aPMMA blend is determined to be miscible in accordance with the literature.¹¹ The glass transition temperature regions (ΔT_{gq}) were calculated as differences between the onset and end points of T_{gq} and are presented in Table I. ΔT_{gq} does not show broadening and this can be taken as a sign for good miscibility between iPMMA and aPMMA. Estimated T_{gq} from the Fox and weight average equations for the iPMMA/aPMMA blend are shown in Table I. It is obvious that the T_g data are fitted well by both equations within experimental error. The results are plotted in Figure 2 and the weight average prediction is shown as the dashed line.

Ternary Blends

The situation for the ternary blends is certainly not as simple as a binary one. Some compositions of the ternary blends of iPMMA, aPMMA, and PSAN showed some

opacity indicating possible immiscibility. The second thermal scans of some compositions of the ternary blends together with iPMMA and PSAN in the same temperature range (as in Figure 1) are shown in Figure 3 for representation. The estimated T_{gsc} and T_{gq} for all the studied ternary blends are listed in Table II. The difference between T_{gsc} and T_{gq} of most compositions in Table II is small. The only exception is the

Table I. Glass transition temperatures of binary blends

	$T_{gsc}/^{\circ}C$	$T_{gq}/^{\circ}C$	T_g^a	T_g^b	$\Delta T_{gq}/^{\circ}C$
iPMMA/aPMMA					
100/0	71.9	74.6	—	—	20
75.0/25.0	80.7, 98.3	82.4	81.2	81.6	14
50.0/50.0	92.8	90.1	88.1	88.7	19
25.0/75.0	94.5	95.6	95.3	95.7	21
0/100	103.4	102.7	—	—	12

^a Estimated from the Fox equation. ^b Estimated from weight average.

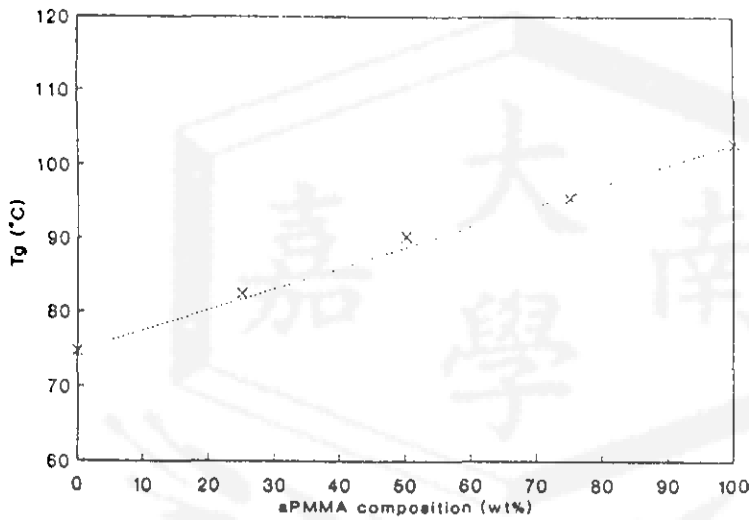


Figure 2. Glass transition temperatures of the iPMMA/aPMMA blend.

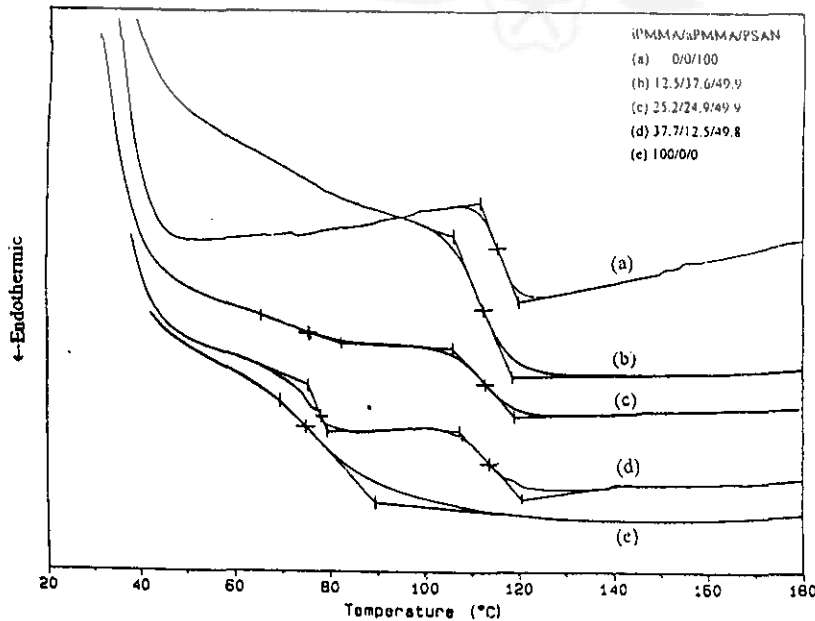


Figure 3. DSC thermograms of the iPMMA/aPMMA/PSAN blend.

Table II. Glass transition temperatures of ternary blends

	$T_{g1}/^{\circ}\text{C}$	$T_{g2}/^{\circ}\text{C}$	T_g^a	T_g^b	$\Delta T_{g1}/^{\circ}\text{C}$
iPMMA/aPMMA/PSAN					
100/0/0	71.9	74.6	—	—	20
56.3/18.7/25.0	72.6, 110.8	73.3, 108.8	89.1	90.0	19, 11
37.5/37.6/24.9	70.9, 107.6	78.1, 107.7	94.5	95.3	16, 12
18.7/56.3/25.0	107.4	105.8	100.0	100.5	12
37.7/12.5/49.8	72.5, 110.0	78.4, 113.9	97.3	98.3	4, 13
25.2/24.9/49.9	72.1, 112.4	76.1, 112.9	101.0	101.8	7, 12
12.5/37.6/49.9	95.1, 112.2	112.7	104.9	105.4	12
18.8/6.3/74.9	111.2	110.6	106.0	106.7	12
12.5/12.4/75.1	109.7	110.5	108.0	108.5	11
6.3/19.2/74.4	114.6	112.9	110.2	110.1	10
0/0/100	117.2	115.1	—	—	9

^a Estimated from the Fox equation. ^b Estimated from weight average.

iPMMA/aPMMA/PSAN (12.5/37.6/49.9) blend which showed one T_g after quenching while before having two T_g s. For the same reason as aforementioned, T_{g1} are used in determining miscibility.

From Table II, the following comments can be made. For the ternary blends with a PSAN composition of ca. 25% or 50%, increasing aPMMA concentration to decrease iPMMA causes the blends to be eventually miscible. The ternary blends with a ca. 75% PSAN are all miscible regardless of the ratio between iPMMA and aPMMA. T_g of the ternaries from either the Fox or weight average equation are tabulated in Table II for reference. For miscible compositions, T_g are always larger than either prediction. For two compositions [iPMMA/aPMMA/PSAN ((12.5/12.4/75.1) and (6.3/19.2/74.4))], weight average predictions are much closer to T_g within 2–3°C.

For the two compositions [iPMMA/aPMMA/PSAN ((56.3/18.7/25.0) and (37.5/37.6/24.9))] the situations are quite similar. They both have two T_g s, one T_g close to iPMMA and the other T_g located between aPMMA and PSAN. Therefore for these two compositions, high T_g phase is composed mainly of aPMMA and PSAN and low T_g phase is rich in iPMMA. A subtle difference can be observed from the T_g s of these two compositions and is stated as follows. Since the PSAN composition is fixed at ca. 25% for both blends, the addition of aPMMA seems to cause low T_g phase to have a higher T_g and little T_g change of high T_g one. Therefore T_g elevation of the low phase is likely due to the incorporation of aPMMA into this phase. The condition for the other two immiscible compositions (composed about half of PSAN and the ratio between iPMMA and aPMMA being ca. 3/1 and 1/1) is almost the same. Based on the T_g data, these two ternary blends seem to separate into two phases composed mainly of iPMMA and PSAN, respectively. aPMMA is distributed in these two phases as a minor component causing a lower T_g of the PSAN-rich phase and T_g elevation of the iPMMA-rich phase.

Phase Diagram

Goh *et al.*⁶ studied similar but somehow different ternary blends. They investigated the miscibility of PSAN/PMMA/PEMA by DSC. PSAN is known to be both miscible with PMMA and PEMA. However,

Table III. Miscibility of ternary blends

PSAN/ca. wt%	25	50	75
aPMMA	+	+	+
iPMMA/aPMMA (ca. 1/3)	+	+	+
iPMMA/aPMMA (ca. 1/1)	—	—	+
iPMMA/aPMMA (ca. 3/1)	—	—	+
iPMMA	—	—	+

+ , miscible; — , immiscible.

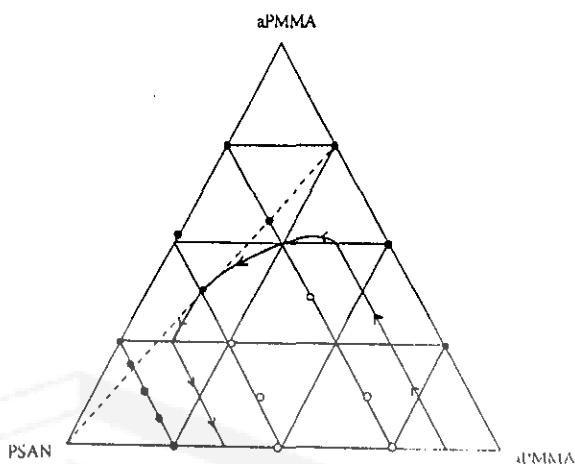


Figure 4. Phase diagram of the iPMMA/aPMMA/PSAN blend. ●, miscible; ○, immiscible.

PMMA and PEMA are not miscible. Therefore PSAN acts as a "cosolvent" between PMMA and PEMA. In their study, they found that the addition of PSAN helps compatibilize the immiscible PMMA/PEMA blends. Our study is different because aPMMA not PSAN is added to homogenize the immiscible pair of PSAN and iPMMA.

The results of Table II and previous data¹⁰ of iPMMA and aPMMA with PSAN were combined and are listed together in Table III. In this table, PSAN composition and the ratio between iPMMA/aPMMA are used as the main indicators. The miscibility is based on the single T_g criterion. The following comments can be drawn from this table. For the ternary blends rich in PSAN compositions (ca. 75%), they are all determined to be miscible. The ternary blends with PSAN concentrations at ca. 25% and 50% and the ratio between iPMMA and aPMMA ≥ 1 are immiscible. It is interesting to note that the ternaries with a 1 to 3 ratio between iPMMA and aPMMA are miscible regardless of PSAN composition (shown as the dashed line in Figure 4). The same results can be seen clearly in a phase diagram. A phase diagram of the ternary blends of iPMMA, aPMMA, and PSAN is shown in Figure 4. A postulated phase boundary was established using limited data points. The boundary curve (the curve with arrows in Figure 4) was connected through middle points between miscibility and immiscibility data and one composition [iPMMA/aPMMA/PSAN (12.5/37.6/49.9)]. This composition can be considered to be borderly miscible because after quenching, two T_g s become one T_g . The boundary curve is rather simplified here but the trend is basically the same even with more data points.

Finally, the reason for ternary blends to be immiscible

is more complicated than binary blends because of more degrees of freedom and more interactions involved. One possible cause for the ternary blends rich in iPMMA to be immiscible arises from interactions between iPMMA and PSAN. The explanation was given previously¹⁰ that iPMMA likely forms a more repulsive interaction with styrene or acrylonitrile than aPMMA and sPMMA.

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REFERENCES

1. T. K. Kwei, H. L. Frisch, W. Radigan, and S. Vogel, *Macromolecules*, **10**, 157 (1977).

2. Y. Y. Wang and S. A. Chen, *Polym. Eng. Sci.*, **21**, 47 (1981).
3. D. Rigby, J. L. Lin, and R. J. Roe, *Macromolecules*, **18**, 2269 (1985).
4. V. Shah, J. D. Keitz, D. R. Paul, and J. W. Barlow, *J. Appl. Polym. Sci.*, **32**, 3863 (1986).
5. J. I. Equizabal, J. J. Irvin, M. Cotazar, and G. M. Guzman, *J. Appl. Polym. Sci.*, **32**, 5945 (1986).
6. S. H. Goh, K. S. Siow, and K. S. Yap, *Thermochim. Acta*, **105**, 191 (1986).
7. S. H. Goh and K. S. Siow, *Thermochim. Acta*, **102**, 281 (1986).
8. W. H. Christiansen, D. R. Paul, and J. W. Barlow, *J. Appl. Polym. Sci.*, **34**, 537 (1987).
9. S. Bicakci and M. Cakmak, *Polymer*, **39**, 4001 (1998).
10. W. P. Hsu, *J. Appl. Polym. Sci.*, accepted.
11. P. E. M. Allen, D. M. Host, V. T. Troung, and D. R. G. Williams, *Eur. Polym. J.*, **21**, 603 (1985).
12. W. P. Hsu and C. F. Yeh, *J. Appl. Polym. Sci.*, in press.
13. F. Fernandez-Martin, I. Fernandez-Pierola, and A. Horta, *J. Polym. Sci. Polym. Phys. Ed.*, **19**, 1353 (1981).
14. T. G. Fox, *J. Appl. Bull. Am. Phys. Soc.*, **1**, 123 (1956).

